An X-ray absorption study on the local structures of highly dispersed supported titanium oxides prepared by a CVD method

Hiroyoshi Kanai ¹ and Hisayoshi Kobayashi

Faculty of Life Science, Kyoto Prefectural University, Shimogamo, Sakyo-ku, Kyoto 606, Japan

Titanium oxides supported on SiO₂ and Al₂O₃ prepared by a CVD method have been characterized by XANES/EXAFS technique. Titanium species on SiO₂ are highly dispersed in a tetrahedral coordination. Titanium species on Al₂O₃ are also highly dispersed in fivefold coordination. High dispersion was substantiated by their photoluminescence emitted by excitation at 300–350 nm.

Keywords: EXAFS/XANES; supported TiO₂; CVD; highly dispersed catalyst

1. Introduction

Many methods have been exploited for characterizing highly dispersed catalysts [1]. One is XAFS analysis which has the advantage that it provides both structural and electronic information [2]. A quantitative analysis of XANES spectra has been successfully applied to the identification of surface transition metal oxide species on supports [3]. There are a few papers on the structures and active species of titanium species in titanium silicalite (TS-1) which catalyzes epoxidation of olefins, ammoximation of cyclohexanone, hydroxylation of benzene [4]. This letter describes highly dispersed phases of supported titanium oxide species which were prepared by a CVD (chemical vapor deposition) technique on the basis of XAFS data.

2. Experimental

Titanium oxide species supported on silica and alumina were prepared by the CVD method. Silica or alumina (SIO-4 and ALO-4 of Reference Catalysts of Catalysis Society of Japan) was evacuated at 673 K for 2 h followed by reacting

¹ To whom correspondence should be addressed.

Ti(OPrⁱ)₄ vapor of saturated pressure at room temperature at 423 K without carrier gas. At this temperature all isopropoxy groups were not decomposed. Isopropoxytitanium species on a support was heated in air at 673 K for 4 h. XAFS samples were prepared as follows: supported titanium oxides were oxidized with 8 kPa of oxygen at 673 K for 2 h and were evacuated at the same temperature for 5 min. Sample powders were sealed into a Q-pack pouch in a dry box. The X-ray absorption experiments were carried out at the beam line 6B and 7C stations of the Photon Factory in the National Laboratory for High Energy Physics (Tsukuba) with a ring energy of 2.5 GeV and a ring current of 150–250 mA. A Si(111) double crystal was used to monochromatize X-rays. The absorption spectra were recorded by using a fluorescence detector at room temperature in the range of photon energies from 4.7 to 6.0 keV for Ti K edge (resolution energy of 0.5 eV at the edge). The data analyses were carried out in a similar manner to those reported previously [3,5].

The phosphorescence spectra were measured at 77 K with a Hitachi 850 fluorophotometer fitted with a phosphorescence attachment. Samples were heated in O_2 (2.7 kPa) at 673 K for 2 h followed by evacuation at the same temperature for 5 min.

3. Results and discussion

3.1, XANES SPECTRA OF Ti K EDGE

Fig. 1 depicts the normalized pre-edge peaks of titanium reference compounds and TiO₂/SiO₂ and TiO₂/Al₂O₃ in dried and hydrated states prepared by the CVD method. The pre-edge peak is assigned to 1s-3d transition. Anatase and rutile TiO₂ in which titanium atoms are in an octahedral symmetry have three small peaks in the pre-edge region. The pre-edge peak of Ti(OPri)4 in a tetrahedral environment is very sharp and high. Two titanium alkoxides have different structures: Ti(OPri)₄ is monomeric and tetrahedral, but Ti(OBun)₄ is an oligomer composed of fivefold coordination [6]. Two shoulder peaks are seen at both sides of a main peak for Ti(OBuⁿ)₄. The high single pre-edge peak of TiO₂/SiO₂(dried) is very similar to that of Ti(OPri)4 which is the precursor reagent. The XANES spectrum resembles that of TiO₂-SiO₂ glass [9]. On hydration the pre-edge peak is shifted to higher energy (1.5 eV) compared to that of the dried sample and is lowered with a shoulder at the higher energy side. The XANES spectrum of TiO₂/Al₂O₃(dried) is similar to that of Ti(OBuⁿ)₄ including the pre-edge peak. When water was adsorbed, the pre-edge peak is shifted to higher energy (0.7 eV), but a small change of XANES spectrum was observed.

XANES spectra were deconvoluted to obtain pre-edge areas (in units of eV) [3] (table 1). Among reference compounds the area of tetrahedral Ti(OPrⁱ)₄ is the largest. The area decreases with increasing coordination number. There is no explicit relation between the pre-edge position and the structure. The area of dried TiO₂/

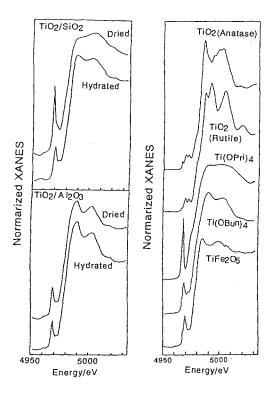


Fig. 1. XANES spectra of Ti-containing reference compounds and TiO/SiO₂ and TiO₂/Al₂O₃ in the dried and hydrated states.

Table 1 Positions and areas of pre-edge peaks

Compound	Pre-edge peak ^a (eV)	Area (eV)		
Ti(OPr ⁱ) ₄	-10.17(0)	1.04	TiO ₄	· · · · · · · · · · · · · · · · · · ·
Ti(OBu ⁿ) ₄	-9.40(0.8)	0.68	TiO ₅	
TiFe ₂ O ₅	-10.47	0.52	TiO ₆	
TiO ₂ (anatase)	-11.92	0.14	TiO ₆	
TiO ₂ (rutile)	-11.27	0.14	TiO ₆	
TiO ₂ /SiO ₂ (dried)	-11.66	1.29		
TiO ₂ /SiO ₂ (hydrated)	-10.1(1.5)	0.42		
TiO ₂ /Al ₂ O ₃ (dried)	-11.5	0.80		
TiO ₂ /Al ₂ O ₃ (hydrated)	-10.84(0.7)	0.47		

^a The peak position is relative to the absorption edge. The number in parentheses is the shift of the peak position on hydration.

 SiO_2 is rather larger than that of tetrahedral $Ti(OPr^i)_4$. Titanium species in TiO_2/SiO_2 should be in tetrahedral coordination. The area decreases upon the adsorption of water parallel to higher coordination.

The pre-edge peak area of $TiO_2/Al_2O_3(dried)$ is between those of tetrahedral and octahedral structures. It is rather close to the area of fivefold coordination compound. The area also decreases with the adsorption of water. The area of $TiO_2/Al_2O_3(H_2O)$ adsorbed) resembles that of $TiFe_2O_5$ in which titanium is in octahedral symmetry [7].

3.2. EXAFS OF Ti K EDGE

The magnitudes of Fourier transforms performed for EXAFS curves in the $3.5-12 \text{ Å}^{-1}$ region are displayed in fig. 2. No further shells are observed except the first shell of the 1-2 Å range which is assigned to Ti-O bonds in dried TiO₂/SiO₂. This should be ascribed to the fact that titanium atoms are well separated from each other and no TiO₂ particles are present. The peaks are centered at lower values than the real distances owing to the phase shift. When water was adsorbed, peaks in the first shell of TiO₂/SiO₂ and TiO₂/Al₂O₃ were broadened indicating that new Ti-O bonds linked to water molecules were formed.

The structural information for Ti-O bonds was derived from inverse Fourier transforms of the peaks in fig. 2 over the range of 0.55-2.15 Å and extracted by using single-scattering theory,

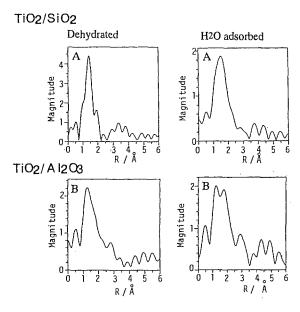


Fig. 2. Fourier transforms of EXAFS oscillations for TiO/SiO₂ and TiO₂/Al₂O₃ in the dried and H₂O-adsorbed states.

$$\chi(k) = \sum_{j} (N_{j}S_{j}/kr_{j}^{2})F_{j}(k) \exp(-2\sigma_{j}^{2}k^{2}) \exp(-2r_{j}/\lambda) \sin[2kr_{j} + \delta_{j}(k)], \quad (1)$$

where N_i is the coordination number of the jth shell at the distance r_i from the Xray absorbing atom, $F_i(k)$ the backscattering amplitude for the photoelectron wave number k, S_i the damping factor for compensation of the loss by inelastic scattering, δ_i the phase shift, σ_i the Debye-Waller factor, and λ the mean free path of the photoelectron (table 2). Tabulated values for $F_i(k)$ and $\delta_i(k)$ by Teo et al. [8] are used for curve-fitting procedures. The reliability of these structural parameters was ascertained with authentic samples of Ti foils and Ti(OPri)₄. The Ti-O bond distance of TiO₂/SiO₂ is 1.77 Å, very similar to that of Ti(OPrⁱ)₄. The bond distance is close to that of TiO_2 – SiO_2 glass (1.79 Å) [9], but differs from those of titanium silicalite (TS-1, 1.87–1.89 Å) [4], titanium oxide (1.85 Å) in TiZSM-5 prepared by treatment of ZSM-5 with TiCl₄ [10] and TiO₂ (anatase, rutile, 1.94 and 1.99 Å). The corrected coordination number of Ti atom based on S_i (eq. (1)) is 3.5 indicating that titanium atoms are highly dispersed in tetrahedral coordination. The environment is very similar to that of TiO₂-SiO₂ glass in the Ti content of 2-10 wt% prepared by flame hydrolysis of SiCl₄ and TiCl₄ vapors [9]. Anatase or an anatase-like structure was obtained for highly dispersed or monolayer titanium oxide on SiO₂ prepared by impregnating of silica into organic solutions of Ti(OPrⁱ)₄ [11]. The structural difference between the two types of TiO₂/SiO₂ from the same precursor is attributed to the difference of the preparation method. The adsorption of water causes a disordered structure from a consideration of the increased Debye-Waller factor, and the additional Ti-O bond which has a longer bond distance (1.93 Å) was assigned to Ti-OH₂.

Although the Fourier transform of TiO_2/Al_2O_3 si not neatly depicted, inverse Fourier transforms were performed over the same region with that of TiO_2/SiO_2 to afford large Debye–Waller factor. The same Ti–O bond distance (1.77 Å) was obtained for TiO_2/Al_2O_3 along with a longer Ti–O bond (1.95 Å). The latter is not assigned to Ti–OH₂ bond, but should be assigned to Ti–O-Al. When water was

Table 2 Curve-fitting for Ti-O shells of supported TiO₂ catalysts

Support	Condition	Structure parameter a			
		R(Å)	CN	$\Delta\sigma^2(10^{-3}\text{Å}^2)$	
SiO ₂	dried	1.77	3.5	7.27	
SiO ₂	hydrated	1.77	3.2	15.0	
	-	1.93	0.63	4.19	
Al ₂ O ₃	dried	1.77	2.9	10.5	
		1.95	0.39	0.20	
Al_2O_3	hydrated	1.78	3.7	14.3	
- +	•	1.98	0.66	0.37	
Ti(OPri)4	in xylene	1.77	4.0	9.82	

^a R: bond length; CN: coordination number; $\Delta \sigma^2$: Debye-Waller factor.

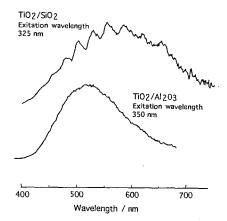


Fig. 3. Phosphorescence spectra of TiO/SiO₂ and TiO₂/Al₂O₃ at 77 K (excitation wavelength 325, 350 nm).

adsorbed, the total coordination number increases including additional Ti-OH₂ bond. The result of XANES/EXAFS data for TiO₂/Al₂O₃ gives the structure composed of fivefold coordination.

Highly dispersed states of TiO₂/SiO₂ and Al₂O₃ prepared by the CVD method were supported from the evidence that both catalysts gave photoluminescence by irradiation of around 300 nm light (fig. 3). The vibrational fine structure of photoluminescence of TiO₂/SiO₂ excited at 325 nm gives an energy separation of 966 cm⁻¹ between (0, 1) and (0, 2) bands which corresponds to the vibration energy of Ti-O bond [12]. There is no vibrational fine structure for TiO₂/Al₂O₃, but the highest peak is shifted to lower wavelength than that of TiO₂/SiO₂.

The CVD technique in which Ti(OPrⁱ)₄ vapor was reacted with OH groups of supports at relatively low temperature is effective for preparing monolayer titanium oxide phases.

The different structure of these two catalysts is reflected in a preliminary test of catalysis for epoxidation of 1-octene with tert-butyl hydroperoxide [13]. High epoxidation activity was obtained for the $\text{TiO}_2/\text{SiO}_2$ catalyst, but no activity was observed for the $\text{TiO}_2/\text{Al}_2\text{O}_3$ catalyst. We may conclude that tetrahedral titanium species on silica is responsible for the active species in the epoxidation of olefins.

Further study on the relationship between structure and catalytic activity is in progress.

Acknowledgement

This work has been performed under the approval of the Photon Factory Program Advisory Committee (Proposal No. 91-002). The authors are indebted to Professor M. Nomura of KEK-PF of Tsukuba in carrying out the X-ray measure-

ments and to Professor S. Yoshida and Dr. T. Tanaka of Kyoto University for help-ful discussions and to Ms. T. Hanada and Mr. H. Yoshida for carrying out experiments. We thank the Data Processing Center of Kyoto University for use of the FACOM M382/780 computer. This work was supported by a Grant-in-Aid for Scientific Research (04650738) from the Ministry of Education, Science and Culture.

References

- [1] K. Tamaru, ed., Surface Science: Application to Theory, Experiments and Catalytic Science (Hyomen no Kagaku) (Gakkai Shuppan Center, Tokyo, 1985).
- J.C.C. Bart, Adv. Catal. 34 (1986) 203;
 J.C.C. Bart and G. Vlaic, Adv. Catal. 35 (1987) 1.
- [3] S. Yoshida, T. Tanaka, T. Hanada, T. Hiraiwa, H. Kanai and T. Funabiki, Catal. Lett. 12 (1992) 277.
- [4] D. Trong On, L. Bonneviot, A. Bittar, A. Sayari and S. Kaliaguine, J. Mol. Catal. 74 (1992) 233, and references therein;
 - D. Trong On, A. Bittar, A. Sayari, S. Kaliaguine and L. Bonneviot, Catal. Lett. 16 (1992) 85.
- [5] H. Kanai, H. Mizutani, T. Tanaka, T. Funabiki, S. Yoshida and M. Takano, J. Mater. Chem. 2 (1992) 703;
 - T. Tanaka, H. Yamashita, R. Tsuchitani, T. Funabiki and S. Yoshida, J. Chem. Soc. Faraday Trans. I 84 (1988) 2987;
 - S. Yoshida and T. Tanaka, Adv. X-ray Chem. Anal. Japan 19 (1988) 97.
- [6] F. Babonneau, S. Doeuff, A. Leaustic, C. Sanchez, C. Cartier and M. Verdaguer, Inorg. Chem. 27 (1988) 3166.
- [7] A.F. Wells, Structural Inorganic Chemistry, 5th Ed. (Clarendon Press, Oxford, 1986) p. 603.
- [8] B.-K. Teo and P.A. Lee, J. Am. Chem. Soc. 101 (1979) 2815;
 B.-K. Teo, EXAFS: Basic Principles and Data Analysis (Springer, Berlin, 1986) pp. 128-133.
- [9] R.B. Greegor, F.W. Lytle, D.R. Sandstrom, J. Wong and P. Schultz, J. Non-Cryst. Solids 55 (1983) 27.
- [10] E. Schultz, C. Ferrini and R. Prins, Proc. 7th Int. Conf. on X-Ray Absorption Fine Structure (1992) p. 490.
- [11] T.M. Salama, T. Tanaka, T. Yamaguchi and K. Tanabe, Surf. Sci. Lett. 227 (1990) L100; K. Asakura, J. Inukai and Y. Iwasawa, Photon Factory Activity Reports No. 7 (1989) 18.
- [12] N.S. McIntyre, K.R. Thompson and W. Weltner Jr., J. Phys. Chem. 75 (1971) 3243; D.R.C. Huybrechts, I. Vaesen, H.X. Li and P.A. Jacobs, Catal. Lett. 8 (1991) 237.
- [13] H. Kanai, S. Imamura and T. Nakai, Abstract on 65th Annual Meeting of Chemical Society of Japan, I (1993) 439.